

Contribution to the study of ZnO films properties prepared by spray pyrolysis

K. DAKHSI^{a,b}, S. ELFARRASS^{b,c}, B. HARTITI^b, M. EBNTOUHAMI^a, A. RIDAH^c, P. THEVENIN^d

^aLaboratory of Electrochemistry Corrosion and Environment, Department of Chemistry, University Ibn Tofail, FS. Kenitra, Morocco

^bLaboratory of Materials Physics and Applications of Renewable Energy, University Hassan II, Morocco FSTM Mohammadia, Morocco

^cLaboratory LEAMCM Faculty of Science ben m'sik, University Hassan II, Morocco

^dLaboratory LMOPS, University of Lorraine, Metz, France

Transparent conducting oxides (TCOs) are widely used as electrodes in optoelectronic devices due to their optical transparency in the visible range and to their electrical conductivity. They can be found in various applications such as photovoltaic solar cells, liquid crystal displays, sensors and organic light emitting devices. Indium tin oxide (ITO) and zinc oxide (ZnO) are the two major TCO materials to be used. In our case, we have deposited Zinc oxide onto pyrex substrates by Spray Pyrolysis technique at temperatures ranging from 350 to 450°C by step of 25°C. The structural properties of the films were studied by X-Ray diffraction (XRD) and confirmed by selected area electron diffraction (SAED) mode. Surface morphology of ZnO thin films was analyzed by scanning electron microscopy (SEM) and transmission electron microscopy (TEM). The films appear to be composed of ZnO nanoparticles (ZnO NPs), whose size increases with temperature. The results showed the presence of the wurtzite phase of ZnO with an orientation along the c-axis of the crystallites. Morphology images of the films obtained by Scanning electron microscopy revealed that pure ZnO film grew as quasi-aligned hexagonal shaped microrods with diameters varying between 100 and 400 nm. UV- visible spectroscopy measurements were also performed to determine the optical band gap.

(Received September 28, 2013; received November 7, 2013)

Keywords: ZnO, Thin films, Spray pyrolysis, XRD, SEM, TEM, UV- visible spectroscopy

1. Introduction

Transparent conductive oxide (TCO) thin films have received more attention from many researchers and scholars because of its industrial optoelectronic applications. Tin-doped indium (ITO) was the most commonly used TCO material because of its excellent optical and electrical properties. But today, it has application problems related to its high cost, toxicity and low stability [1]. Zinc oxide (ZnO) has therefore attracted considerable attention on account of its good electrical and optical properties in combination with its abundance in nature and its non toxicity and can also be taken as promising alternative candidate to ITO [2].

ZnO is an II-VI group semiconductor material with a hexagonal wurtzite crystal structure. It has broader direct band gap energy at 3.37 eV and a higher excitation binding energy at about 60 meV under room temperature. With these notable physical characteristics and low cost, it has broader applications such as in gas sensors, surface acoustic wave components and short-wavelength light-emitting materials [2].

Various deposition techniques have been widely used to produce ZnO thin films. However, seeking the most reliable and economic deposition technique is the main goal. The most studied techniques include thermal evaporation [3, 4], sputtering [5, 6], chemical vapor deposition [7, 8], sol-gel [9, 10], pulsed laser deposition [11,12]. Electrochemical deposition [13,14] and Spray

Pyrolysis [15,16]. Among these techniques, spray pyrolysis is one of the most widely used methods because of its many advantages such as simplicity, low cost, high purity, excellent control of chemical uniformity and stoichiometry in multi-component system. It can be performed at atmospheric pressure and does not require any vacuum set up. Very common compounds can be used as starting material, such as zinc nitrate [17], zinc acetate [17, 18], or zinc chloride [19].

Another major advantage of spray pyrolysis is that it can achieve excellent thickness uniformity of the films, and also can be easily adapted for production on large-area substrates.

In this study, ZnO thin films were deposited by spray pyrolysis on Pyrex substrates starting with an aqueous solution containing zinc chloride. The influence of substrate temperature on the structural, morphological and optical properties of ZnO thin films was investigated without any post annealing.

2. Experimental details

2.1 Deposition conditions

The depositions were performed onto Pyrex substrates, which were previously ultrasonically cleaned and kept at temperatures from 350 °C to 450 °C by step of 25°C during the deposition. The starting solution is

when the substrate temperature increases. This may be due to the more evaporation of the solvent on the substrate at high temperatures giving place to very thin films.

Table 1. Variation of film thickness, crystallite size strain, stress and lattice parameters with substrate temperatures

Temperature	crystallite size (nm)	thickness (nm)	c(Å)	a(Å)	epsilon z	sigma (Gpa)
T350°C	33,44	380,00	5,21	3,26	0,10	2074,26
T375°C	37,24	315,00	5,22	3,26	0,16	3506,48
T400°C	44,42	250,00	5,22	3,25	0,18	3843,96
T425°C	45,05	189,00	5,21	3,25	0,08	1670,93
T450°C	42,31	166,00	5,20	3,25	-0,04	-831,35

The lattice constants c and a of the ZnO films are in agreement with standard value from JCPDS-ASTM No .01-1136 $c_0= 5.21\text{Å}$ and $a_0= 3.242\text{ Å}$. They are calculated using the following equation: [21].

$$d_{hkl} = \frac{a}{\sqrt{\frac{4}{3}(h^2 + k^2 + hk) + l^2 \frac{a^2}{c^2}}} \quad (2)$$

The average uniform strain (epsilon z) in the lattice along the c-axis has been estimated using the following expression:

$$\varepsilon_z = \frac{c - c_0}{c_0} \times 100\% \quad (3)$$

The stress (σ) in the plane of the film can be calculated using the biaxial strain model: [22]

$$\sigma_{film} = \frac{2C_{13}^2 - C_{33}(C_{11} + C_{12})}{2C_{13}} \varepsilon_z \quad (4)$$

Where $C_{11}=209.7\text{GPa}$, $C_{12}=121.1\text{GPa}$, $C_{13}=105.1\text{GPa}$, and $C_{33}= 210.9\text{ GPa}$ are the elastic stiffness constants of bulk ZnO. Minimum stress was observed at Temperature 425°C, due to higher orientation of crystallites along the c-axis, The negative values of strain for the films indicate that the lattice constant c is compressed, as compared to the unstressed powder sample.

3.2 Morphological characterization

Fig.4 shows the SEM images of samples elaborated at 350°C, 375°C, 400°C, 425°C and 450°C. For all the deposited films, one can see nanocrystals with densely packed grains distributed evenly throughout the entire surface of the film. The grain size increases with increasing the substrate temperature as shown in table 1. These images show clearly a compact and homogenous surface for films deposited at all temperatures. The films are very uniform, with a dense surface without visible holes or faulty zones.

The EDAX analysis showed that all the layers are stoichiometric (50% of Zn and 50% of O) (Fig. 4).

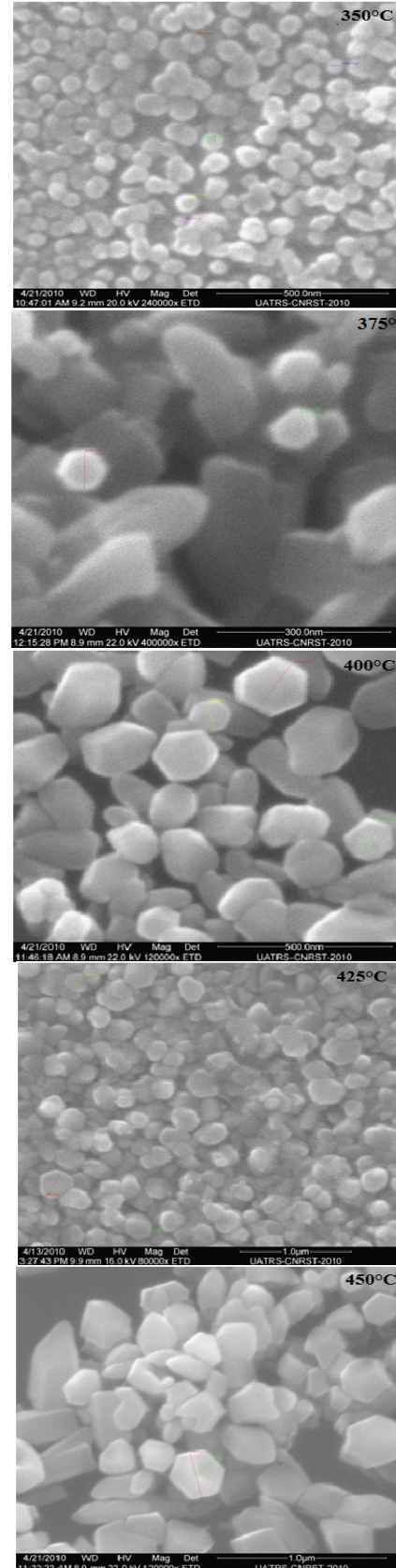


Fig.4. SEM images of ZnO thin films synthesized at different substrate temperatures

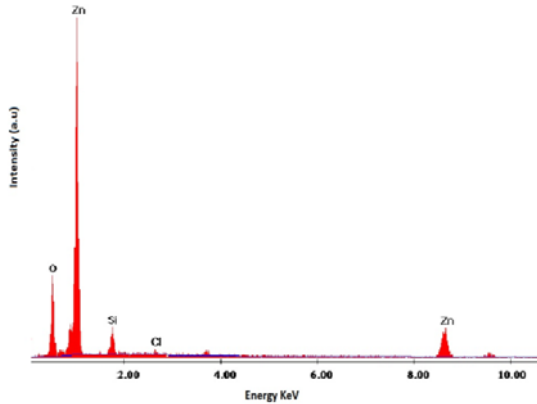


Fig.5. EDAX analysis spectrum of the ZnO thin film.

3.3 Optical properties

Fig.6 displays the transmittance spectra of ZnO thin films deposited onto Pyrex substrates by Spray pyrolysis at different substrate temperatures. The transmissions in the visible range (400nm-800nm) reach values of the order of 90%. In a general way, an increase in substrate temperature improved the transmission as can be observed on the deposits performed at temperature 350°C, 375°C and 450°C in comparison to those prepared at 400°C and 425 °C.

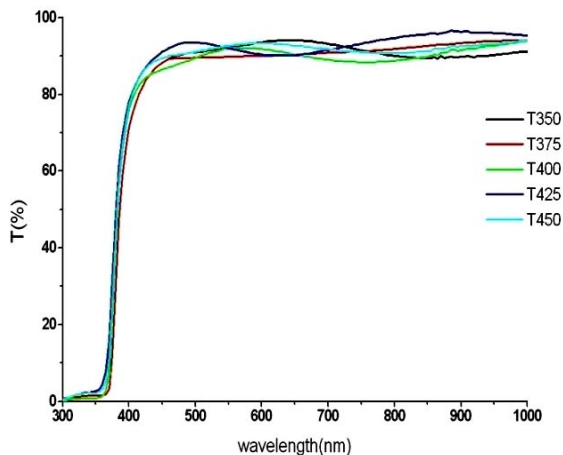


Fig.6. Transmission spectra of ZnO thin films.

A sharp absorption edge can be observed at about 375 nm, which corresponds to an energy gap of 3.305 eV, as determined for the film deposited at 425°C presented in Fig.7. And it can be observed that this improvement can be attributed to either the decrease in thickness or the improvement in structural homogeneity and crystalline quality.

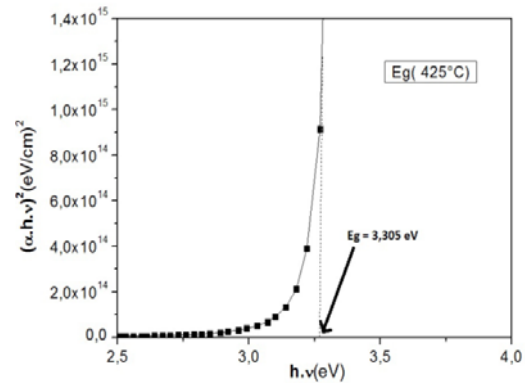


Fig.7. Variation of $(\alpha \cdot h \cdot \nu)^2$ as a function of $h \cdot \nu$ of ZnO thin film elaborated at 425°C

In Fig. 7, one can see the variation of $(\alpha \cdot h \cdot \nu)^2$ as a function of $h \cdot \nu$ of ZnO thin film elaborated at 425°C since it is the best elaboration temperature. The absorption coefficient α is calculated from the relation [23].

$$T = (1 - R) e^{-\alpha t} \quad (5)$$

Where T is the transmittance, R is the reflectance and t is the film thickness. The optical band gap of film is determined by applying the Tauc model in the high absorption region.

$$\alpha h \nu = A \sqrt{h \nu - E_g} \quad (6)$$

Where $h \nu$ is the photon energy, E_g the optical band gap, and A is a constant. The band gap of the film has been calculated using Tauc's plot of $(\alpha \cdot h \cdot \nu)^2$ versus $h \cdot \nu$ as shown in Fig.7 and by extrapolating the linear part of the absorption edge to find the intercept with energy axis. A band gap of 3.305 eV has been obtained for the film deposited at 425°C.

4. Conclusion

In this study, we have deposited ZnO thin films by spray pyrolysis technique on a Pyrex substrate at different temperatures using Zinc chloride dissolved in deionised water. The results showed that the deposited ZnO thin films are uniform. The structural analysis of X-rays diffraction and TEM analysis showed that the films deposited are of wurtzite structure (phase zincite, JCPDS 01-1336) with a preferential orientation along the [002] direction. The study of the microscopic structure at different temperatures revealed a homogeneous and uniform grain structure. In the other case Particle size increases with increasing substrate temperature from 350 °C to 450 °C. The transmittance spectra show that the films prepared at 400°C and 425°C exhibit a transmission higher than 90% in the visible region with a sharp

fundamental absorption edge at about 375 nm, corresponding to gap energy of 3.305 eV. The films prepared at temperature 350°C, 375°C and 450°C have a low transmission, and those prepared at 400°C 425 °C have a higher one. It can be observed that, in general, an increase in substrate temperature improved the transmission. This improvement can be attributed to either the decrease in thickness or the improvement in structural homogeneity and crystalline quality.

Acknowledgments

This work was partially funded by HORIZON project funded by AUF under contract number 59113PS019; the authors thank researchers from LMOPS (Metz, France), IES (Montpellier, France) and GOPS (Valencia, Spain) laboratories for their help during stay of LPMAER researchers.

References

- [1] Y. K. Tseng, G. J. Gao, S. C. Chien, *Thin Solid Films*, **518**(22), 6259 (2010).
- [2] B.Y. Oh, M.C. Jeong, W. Lee, J.M. Myoung, *J. Cryst. Growth* **274**, 453 (2005).
- [3] J. Ma, F. Ji, D. Zhang, H. Ma, S. Li, *Thin Solid Films* **357**, 98 (1999).
- [4] S. A. Aly, N.Z. El Sayed, M. A. Kaid, *Vacuum* **61**, 1 (2001).
- [5] T. Minami, Y. Ohtani, T. Miyata, T. Kuboi, *J. Vac. Sci. Technol. A* **25**, 1172 (2007).
- [6] J. G. Lu, T. Kawaharamura, H Nishinaka, Y. Kamada, T. Ohshima, S. Fujita, *J. Cryst Growth* **299**, 1 (2007).
- [7] A. Martin, J.P. Espinos, A. Justo, J.P. Holgado, F. Yubero, A.R.G. Elipe. *Surf. Coat. Technol.* **151**, 289 (2002).
- [8] V. Musat, B. Teixeira, E. Fortunato, R.C.C. Monteiro, *Thin Solid Films* **502**, 219 (2006).
- [9] H. M. Zhou, D.Q. Yi, Z. M. Yu, L.R. Xiao, J. Li, *Thin Solid Films* **515**, 6909 (2007).
- [10] Y. Liu, L. Zhao, J. Lian, *Vacuum* **81**, 18 (2006).
- [11] N. Fathy, M. Ichimura, *J. Cryst. Growth* **294**, 191 (2006).
- [12] Y. Gao, F. Nagai, Y. Masuda, F. Sato, K. Koumoto, *J. Cryst. Growth* **286**, 445 (2006).
- [13] Y. Takashige, T. Nebiki, T. Narusawa, *Solid State Phenom.* **135**, 124 (2007).
- [14] P. Singh, A. Kumar, D. Deepak Kaur, *J. Cryst. Growth* **306**, 303 (2007).
- [15] N. Nakayama, K. Ito, *Appl Surf Sci*, **92**, 171 (1996).
- [16] N. Kamoun, H. Bouzouita, *Thin Solid Films* **515**, 5949 (2007).
- [17] M. Krunk, E. Mellikov, *Thin Solid Films* **270**, 33 (1995).
- [18] B. J. L. Lokhande, M.D Uplane, *Appl. Surf. Sci.* **167**, 243 (2000).
- [19] J. Zhao, X. M. Li, J.M. Bian, W. D Yu., C. Y. Zhang, *Thin Solid Films*, **515**, 1763 (2006).
- [20] T. V. Vimalkumar, N. Poornima, C. Sudha Kartha, K. P. Vijayakumar. *Applied Surface Science* **256**(20), 6025 (2010).
- [21] T. Prasada Rao, M.C. Santhosh kumar. *Appl.Surf.Sci*, **255**(8), 4579 (2009).
- [22] Tse-Chang Li, Po-Tsung Hsieh, Jen-Fin Lin. *Ceramics International*, **37**(7), 2467 (2011)
- [23] E. Rudigier, B. Barcones, *J. Appl. Phys.* **95**, 5153 (2004).

*Corresponding author: dkhissi345@hotmail.com